Heterogeneous integration and precise alignment of InP-based photonic crystal lasers to complementary metal-oxide semiconductor fabricated silicon-on-insulator wire waveguides

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The integration of two-dimensional III-V InP-based photonic crystal and silicon wire waveguides is achieved through an accurate alignment of the two optical levels using mix-and-match deep ultraviolet (DUV)/electron beam lithography. The adhesively bonded structures exhibit an enhancement of light emission at frequencies where low group velocity modes of the photonic crystal line defect waveguides occur. Pulsed laser operation is obtained from these modes at room temperature under optical pumping. The laser light is coupled out of the Si waveguide via grating couplers directly to single mode fiber. © *2010 American Institute of Physics.*
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I. INTRODUCTION

In the quest for efficiency, integrated optics has recently turned its attention to development of subwavelength, high contrast photonic components. Preferably based upon a silicon platform, these would allow one to envision low-cost and high component density.[1] To achieve these ambitious goals Si technology takes advantage of processing know-how from the electronics industry. Mature complementary metal-oxide semiconductor (CMOS) processing technology renders silicon unsurpassed in terms of fabricated device quality. Furthermore, the ability of low-loss Si wire waveguides to be bent with tight radii of curvature predisposes them to small device footprints and thus large scale integration. Beyond purely passive features such as guiding and filtering, the inclusion of active components in the Si platform would vastly enhance the portfolio of optical functions. For this we require devices capable of emitting, modulating, and detecting light, ideally with low power expenditure. Heterogeneous integration of III-V materials onto silicon could offer an efficient solution, providing tailor-made optoelectronic properties. A wide variety of materials can be bonded, to allow mixing and matching of different functionalities.[2] Using this technology Fabry–Perot, distributed feedback and mode-locked lasers have been integrated with silicon-on-insulator (SOI) rib waveguides and integrated microdisk lasers coupled to SOI wire waveguides have been demonstrated.[3] Patterning this active material on the wavelength scale also permits control of its photonic properties. In this context two-dimensional (2D) photonic crystals (PhCs) allow us to achieve ultrasmall components and can be configured to substantially augment the light-matter interaction, thus reducing laser thresholds and switching energies.[4] Indeed carrier nonlinearities in patterned III-V materials, [more specifically InP-based Quantum Wells (QWs) Refs. 5 and 6], have larger magnitudes and faster response times compared to silicon. Monolithic integration of active PhC structures with passive waveguides has been investigated using regrowth of passive material[7] and also intermixing to produce all-InP-based alternatives.[8] In terms of PhCs which have been integrated onto Si, in Ref. 9 high contrast modulation was achieved using a benzocyclobutene (BCB) bonded InP-based surface emitting PhC. Similar graphite lattice PhC structures were also integrated onto a Si wafer using AuIn eutectic bonding, and continuous wave (CW) laser operation[10] was then established due to the five-fold higher thermal conductivity of the SiO_2 cladding. Rapid modulation of these PhC lasers integrated on Si was demonstrated by dynamic measurements of the pulse turn on/off.[11] These demonstrations point the way to all-guided hybrid systems, which is our concern here. The enhancement of silicon photonics by III-V PhC based optical functions would make for a versatile integrated photonics platform.

In this work, we study the integration of a PhC membrane containing InGaAsP/InGaAs QW material and SOI wire waveguides, using die-to-die bonding (as shown schematically in Fig. 1). For this heterogeneous integration, we choose adhesive bonding as it is a rapid, convenient alternative to molecular wafer bonding.[12] It is desirable that the high resolution PhC patterning that we subsequently apply to the InP slab has a close spatial registration with the underlying Si waveguide.

In Sec. II, we discuss the proposed hybrid platform consisting of InP-based membrane material bonded; patterned with a 2D PhC structure; and coupled to SOI photonic wires. In Sec. III, the technological procedure for the fabrication of the hybrid structures is detailed. Sec. IV shows the measurements of the alignment accuracy between the two optical

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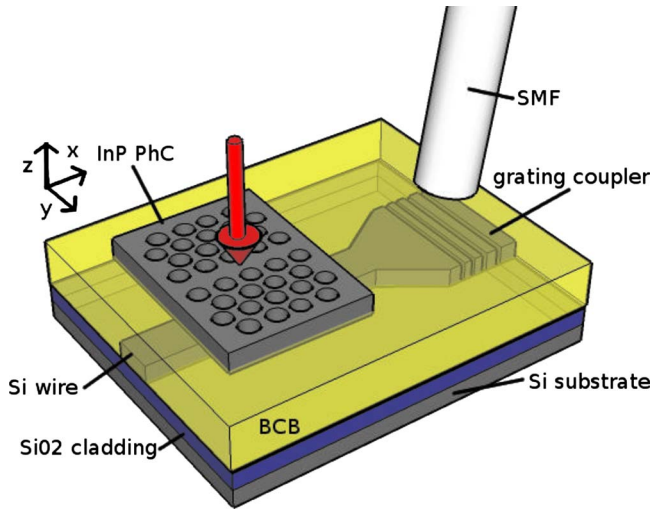c)Also at Université Paris 7 Denis Diderot, Paris, France.

FIG. 1. (Color online) Hybrid III-V PhC/SOI wire structure and experimental configuration. Our PhC defect waveguide hybrid structure is optically surface pumped, as indicated by the red arrow. The underlying Si wire waveguide captures the laser emission which is then diffracted by a grating coupler and collected by a cleaved SMF.

levels of our structures. Finally, in Sec. V we report the optical measurements of the device emission. The luminescence experiments are compared to numerical simulation and results on laser emission are presented.

II. HYBRID PLATFORM: III-V 2D PHC BONDED AND COUPLED TO SOI WIRE

SOI photonics takes advantage of silicon's natural oxide as a waveguide cladding. Wire waveguides with cross sectional dimensions below approximately 220 high by 500 nm wide, provide tight confinement of the optical field and can be seen [in Fig. 2(a)] to be single mode (per polarization). We denote the polarization of the wire modes following the dominant components of their fields, based upon those of the

SOI slab waveguide. TE-like modes are dominated by components of the electric field in the plane of the slab and TM-like with the magnetic field in the plane of the slab. With asymmetry in the vertical direction the wire modes are hybrid, and so here we only loosely apply the terms TE/TM-like, to indicate the dominant polarization.

PhCs are high contrast periodic dielectric structures, which promote strong interference between forward and backward traveling optical waves. Operating at either the band edge of a uniform PhC lattice or at the mode edge of a defect waveguide mode gives control over the spatial localization of the optical field and concurrently, control of its group velocity.[13] We choose a defect waveguide configuration, which is obtained by suppressing one row of air holes in the triangular PhC lattice (W1), in order to provide a system, which is ostensibly monomode (one mode per polarization per waveguide) and highly confined.

In Fig. 2(b) the three-dimensional (3D) band structure of a W1 waveguide is plotted, calculated using a guided mode expansion along the Γ-K direction.[14] Modes of both polarizations (TE-like and TM-like) are plotted and these are displayed in blue for odd and in red for even lateral symmetry. Mirror reflection with respect to the vertical plane (x-z) is a symmetry operation of the waveguide. The modes are classified as even ($\sigma_{xz}=+1$) or odd ($\sigma_{xz}=-1$) with respect to this mirror plane. The dispersion of the odd TE-like defect mode close to (k=π/a, u=a/λ=0.275), is especially flat. In this low v_g regime the interaction between the matter and the optical field can be greatly enhanced.[15] We would like to interact efficiently with the guided PhC modes in the window k_x=0.45−0.55×2π/a for normalized frequencies, u =0.275−0.3, beneath the BCB light line. This is a regime that previous waveguide—PhC coupling schemes have been unable to attain due to their lower effective waveguide mode indices.[16] We therefore opt for a nominal SOI wire width of 300 nm, this is near the lower limit of the patterning capa-
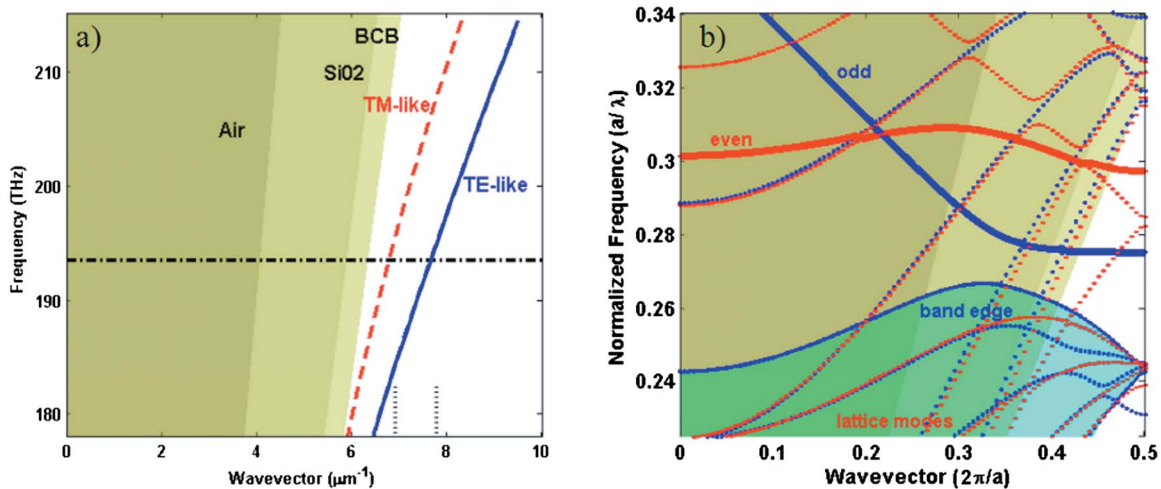

FIG. 2. (Color online) Dispersion relations. (a) The TE-like (solid line) and TM-like (dashed line) modes of a 220×300 nm^2 Si waveguide with a SiO$_2$ substrate, and BCB superstrate (see inset to Fig. 3). The light cones of the various cladding materials, air, SiO$_2$, and BCB are denoted by shading. The vertical dotted lines mark the range of lithographic tuning in the experiment. The horizontal line marks the peak gain wavelength of our QWs. (b) The 3D PhC bandstructure calculated using a guided mode expansion for an InP membrane with a BCB substrate and air superstrate (with air holes). The thick line (marked odd) corresponds to the odd TE-like defect mode and the thick line (marked even) the even TE-like defect mode. The TE-like band edge modes are shaded. The TM-like lattice modes are plotted as dotted lines. The period is a=435 nm, hole radii r=0.3 a and the height of the InP is 255 nm, the supercell is 5 $\sqrt{3}$ a wide.
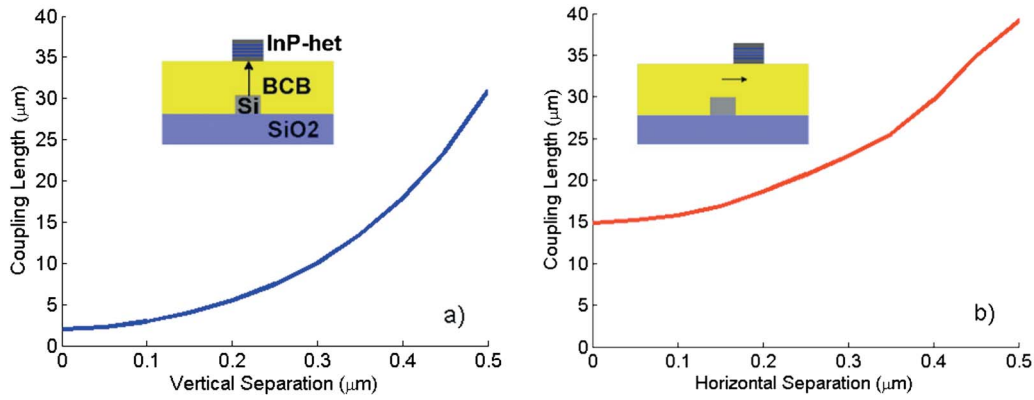
FIG. 3. (Color online) The variation in coupling length of an InP-based (255×360 nm^2) wire phase matched to a (220×300 nm^2) Si wire as a function of the (a) vertical and (b) lateral offset between the guides at a constant vertical offset of 400 nm for $\lambda = 1.55$ μm.

bilities of the DUV process. The dispersion of a 300 nm wide waveguide is plotted in Fig. 2(a). One TE-like and one TM-like mode are guided. For a PhC period of a=435 nm and r=0.3a, the mode edge of the InP W1 defect waveguide occurs at 1.55 μm, and has an effective index, n_{eff} of 1.78. At the same wavelength a 300 nm Si guide sitting on a SiO$_2$ substrate with BCB superstrate has an n_{eff} of 1.91. By fabricating a series of structures with different periods we can tune the modes of the coupled waveguides and investigate different coupling regimes at or close to a phase matched condition.

A couple of drawbacks naturally exist. First heterogeneous integration creates a vertical asymmetry, which permits polarization mixing. Odd (even) TE-like modes couple with odd (even) TM-like modes, respectively, and any lateral misalignment removes the symmetry entirely. Second, the relatively high refractive index of the BCB (1.54) cladding, as compared to SiO$_2$ (1.45) or air (1) reduces the optical confinement [as shaded in Figs. 2(a) and 2(b)]. Although the layer thickness of the BCB is in the range of hundreds of nanometers its refractive index fixes the limit to lossless guiding in this system. Modes folded above the light line of the BCB are coupled to the radiation continuum giving a bandwidth limit to the devices. Both PhC and SOI waveguides offer strong modal confinement and so we need to be able to appreciate the tolerance limits of our alignment techniques, if we want to transfer energy between the two waveguides.

With the aim of understanding the alignment tolerance we briefly investigate, using a numerical calculation, the coupling between two index matched waveguide modes and its dependence upon vertical and lateral misalignment. This study neglects the periodicity of our experimental waveguide system. Using a finite difference fully vectorial mode solver, we first find the parameters which phase match the fundamental mode of a 220×300 nm^2 SOI waveguide (SiO$_2$ substrate and BCB superstrate) with that of a 255 nm high InP/InGaAs QW waveguide (BCB substrate and air superstrate), each having an effective index of $n_{eff}=1.91$. The coupling length increases exponentially with separation in both z and y directions, as expected. It can be seen [from the data presented in Fig. 3(b)] that tight lateral positioning control is required on a scale of about 100 nm (which is only possible

using e-beam alignment), 100 nm misalignment would give an increase in the coupling length of about 15%.

III. FABRICATION

A. Si waveguide definition and InP wafer growth

The SOI waveguides are patterned using 193 nm DUV lithography on a CMOS line. These are then etched with a ClHBr reactive ion etch (RIE). The 220 nm Si layer is entirely etched through, leaving a 220×300 nm^2 waveguide on a 2 μm buried oxide, with a silicon substrate. Alignment marks for subsequent processing are defined in the Si on the same mask layer and relative to the waveguide structures. The Si wires are tapered up to a width of 10 μm for output coupling via shallow etched grating couplers.[17] The InP material is grown by metal-organic chemical-vapor deposition, and contains four InGaAs/InGaAsP QWs emitting at a peak photoluminescence (PL) wavelength of 1.53 μm.

The silicon waveguide dimensions have been confirmed by scanning electron microscope (SEM) measurements of the width (326 nm), the wire width being varied deliberately across the 8 in. wafer following the exposure dose. The III-V slab thickness (255 nm) was measured by wet etching back the InP and QW layers and using a Dektak surface profiler.

B. Bonding

A 1.4×1 cm^2 SOI die is diced from an 8 in. wafer and a roughly 5 mm^2 die of InP is cleaved [see Fig. 4(a), (i)]. Achieving a high quality bonded interface requires two clean surfaces. A standard RCA-1 (DI water+NH$_4$OH+H$_2$O$_2$, 5:1:1 at 70 °C) clean is performed for the Si layer and the InP is cleaned with acetone-IPA. The BCB is diluted in an appropriate ratio of mesitylene in order to obtain the chosen spun film thickness,[18] which for the samples fabricated here is ~400 nm (top of Si to lower InP surface). After application of an adhesion promoter (AP3000) the BCB is spun (5000 rpm, 30 s) onto the SOI, outgassed at 80 °C for several minutes to avoid the presence of bubbles that may otherwise introduce defects into the final structure. Finally the InP material is gently placed over the site of the patterned waveguides, while the sample is heated to 150 °C [see Fig. 4(a), (ii)]. In order to polymerize the BCB and finalize the
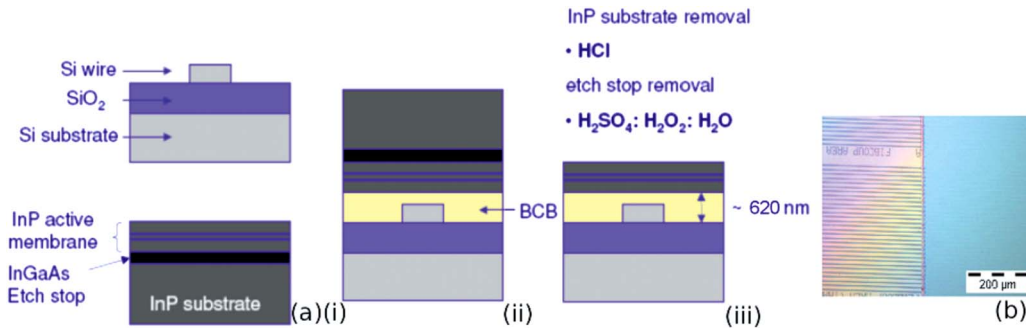
FIG. 4. (Color online) (a) Steps to achieve membrane bonded above waveguide. The two wafers are cleaned, then diluted BCB is spun onto the SOI waveguides and the InP die is positioned on top. After hard baking, the InP substrate is removed with a wet etch. (b) Optical image of SOI waveguides with bonded InP slab on the right hand side, the Si waveguides are just visible beneath the InP.

bonding a hard bake is performed in a nitrogen atmosphere, for 2 h at 300 °C. Once hard baked the InP substrate is removed by HCl wet etch. The InGaAs etch stop is removed using $H_2SO_4:H_2O_2:H_2O(3:1:1)$ to leave the 255 nm InP-based membrane bonded to the SOI [see Fig. 4(a), (iii)]. An optical image of the bonded material is shown in Fig. 4(b).

C. PhC patterning and alignment

Patterning the PhCs into the bonded InP material requires a two stage masking process. A 300 nm Si_3N_4 mask is deposited by plasma-enhanced chemical-vapor deposition to act as a hard mask. This layer also serves to protect the BCB and underlying silicon guides in later processing steps. On top of this we spin coat and bake a 450 nm polymethyl methacrylate (PMMA) mask.

To align our guides, our "semiautomatic" alignment technique requires four marks defined in the same high resolution mask layer as the Si waveguides. In order for the Leica EPBG5000+e-beam writer to be able to detect and recognize these marks, they must be of suitable contrast. The backscattered signal is directly related to the atomic number of the material in which the mark is defined. The abruptness of the edges of the mark is paramount to ensuring the correct measurement of the position of their center. In these thin semiconductor films, at the high electron writing energy (100 keV) employed, the backscattered signal is characteristically weak and of poor contrast. To add to this difficulty the multilayer semiconductor/dielectric stack is prone to charging, as the BCB (and also the SiO_2) is a very efficient insulating layer. To overcome these difficulties a wide range of mark geometries and configurations was tested. Our final choice consists in using 10×10 μm^2 squares etched through the 220 nm silicon and the 2 μm SiO_2 layers. The contrast of these CMOS compatible marks allows us to repeatably detect them and thereby align the PhC to the Si waveguides. The marks permit us to correct for the rotation, shift, and scaling between the Si and InP layers. The PhC waveguide patterns are exposed in the PMMA along with a 2 μm wide moat (in the same high resolution layer).

The moat allows us to very simply discard the superfluous InP at a later processing step, avoiding the absorption that this material would otherwise cause. Following the development of the PMMA, the pattern is transferred into the Si_3N_4 mask by a RIE using a SF_6/CHF_3 chemistry [gas ratio 10/14 SCCM (SCCM denotes cubic centimeter per minute at STP)]. To avoid reflow during the subsequent high temperatures steps the PMMA is then also removed in a RIE dry etch organic cleaning process. The PhC is etched into the InP layer by inductively coupled plasma (ICP)-RIE (Ref. 19) with Cl_2/BCl_3N_2 chemistry. The Si_3N_4 mask is then removed by repeating the previous RIE process. We then remove the superfluous InP material. Using the alignment marks and a negative resist (maN 2410) a low resolution area is exposed, using e-beam lithography, 1 μm narrower than the moat. This mesa protects the underlying PhC from a three step wet etching cycle using HCl, $H_2SO_4:H_2O_2:H_2O(3:1:1)$ and once again HCl, which removes the InP and the InGaAs/InGaAsP QWs. Samples of two different lengths, 30 and 200 periods, were fabricated. The technological steps as well as the optical images of the sample after each step are shown in Figs. 5 and 6.

IV. ALIGNMENT OF WAVEGUIDES

During the InP etching step, the BCB layer was slightly thinned and actually allow us to obtain reasonably contrasted SEM images of the ends of the PhCs and the silicon waveguide layer, beneath the residual BCB, as can be seen in Fig. 7.

By fitting the waveguide position and the hole centers over a selection of guides, we can gain a statistical measure of our alignment capabilities.

To determine the position of the waveguide at the end of the PhC an adaptive supergaussian function (variable width, height, and center) is fitted to each transverse section of the wire, (determining the position of the guide provides the largest measurement error, WG error, as displayed in the 3rd
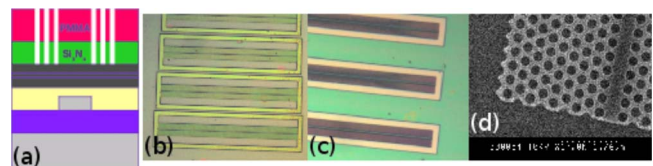

FIG. 5. (Color online) (a) The hard Si_3N_4 mask is etched using Fluorine-based RIE through the patterned PMMA obtained after aligned electron beam lithography. (b) Optical images showing the PhC patterning of first the soft PMMA mask (aligned with the guide below) and (c) its subsequent transfer into the Si_3N_4 hard mask by RIE etch. (d) An SEM image reveals the island of patterned Si_3N_4.
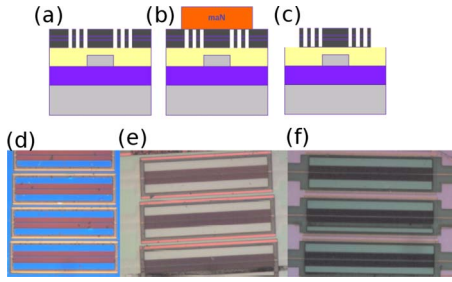
FIG. 6. (Color online) (d) The Si_3N_4 mask (a) The InP layer is etched using inductively coupled plasma (ICP), and in the lower optical image (e) the Si waveguides become visible. The Si_3N_4 mask is removed and a electronic negative resist maN is patterned (b) to efficiently protect the PhC while the superfluous InP material is removed. (c) and (f) The final structures: bonded, accurately aligned, and patterned on top of the Si wire waveguides.

TABLE I. The fitted alignment offset tolerances for nine ~90 μm PhC long waveguides, along with the error in determining the waveguide position, WG error.

	Left hand side		Right hand side	
	Alignment offset (nm)	WG error (nm)	Alignment offset (nm)	WG error (nm)
Guide 1	−21	7	−34	5
Guide 2	16	8	15	3
Guide 3	−7	5	12	3
Guide 4	−32	3	14	4
Guide 5	18	4	21	3
Guide 6	−57	4	−80	4
Guide 7	16	6	44	3
Guide 8	3	4	33	3
Guide 9	−18	4	28	3

and 5th columns of Table I). A linear fit through these center positions is extrapolated to the end of the PhC. After thresholding the PhC region the edges of all the shapes (~30 holes) in the structure are detected, using a custom-written code. Each promising candidate is fitted with a circle using Taubin's method as shown in Fig. 8. The circle center coordinates are then least-squares fitted to the designed structure (~1 nm error). The predicted end position of the PhC is then subtracted from the waveguide position, to give the alignment offset. These results are displayed in Table I.

We checked each of the nine individual longer waveguides to determine a statistical alignment accuracy. Averaging over the alignment errors for left and right hand side ends of the PhC waveguides gives 26 ± 19 nm. This accuracy is well below the 100 nm value which we saw (in Fig. 3) would lead to an increase in the coupling length of 15%. With good control over the fabricated waveguides we now look at the optical properties of these hybrid structures, first numerically and then optically.

V. MODELING, OPTICAL CHARACTERIZATION, AND DISCUSSION

A. Numerical simulations

In order to study numerically the coupled system we have chosen a 3D finite difference time domain (FDTD) (full-vectorial) modeling tool.[20] We consider the system to be entirely passive, i.e., we do not include the absorption/gain of the QWs. In the propagation direction x, the InP PhC has discrete translation symmetry whereas the Si wire has continuous translation symmetry, so the coupled system adopts the period, a, of the PhC waveguide. We, therefore, create a supercell and apply Bloch boundary conditions in the x direction and absorbing boundary conditions in the lateral (y) and vertical (z) directions. We can (given our high, measured alignment accuracy, which is approximately equal to one FDTD grid step) apply an odd (or even) symmetry plane at y=0.

In Fig. 9(a) the coupled bandstructure is presented, calculated for an InP PhC W1 waveguide (period, a=455 nm)
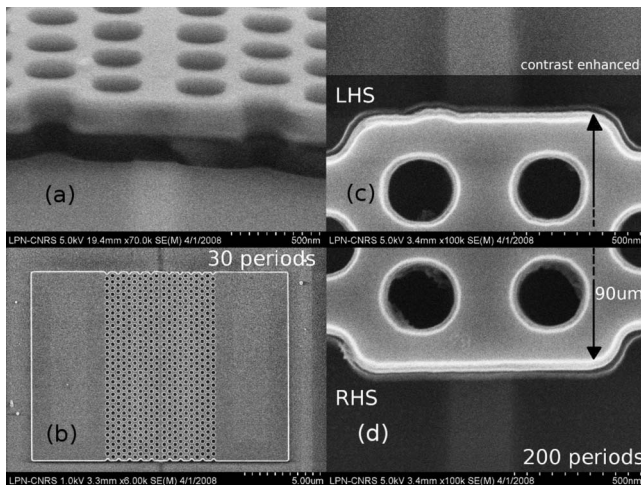

FIG. 7. (a) SEM angled view of the end of the sample showing the patterned InP layer on top of the etched BCB, the Si guide is barely visible to the SEM, here we see a slightly lighter region, (b) full InP mesa containing etched ~13 μm "short" PhC (c) LHS of longer guide with enhanced contrast to display alignment with underlying waveguide, (d) RHS of waveguide, which is ~90 μm further along the waveguide.
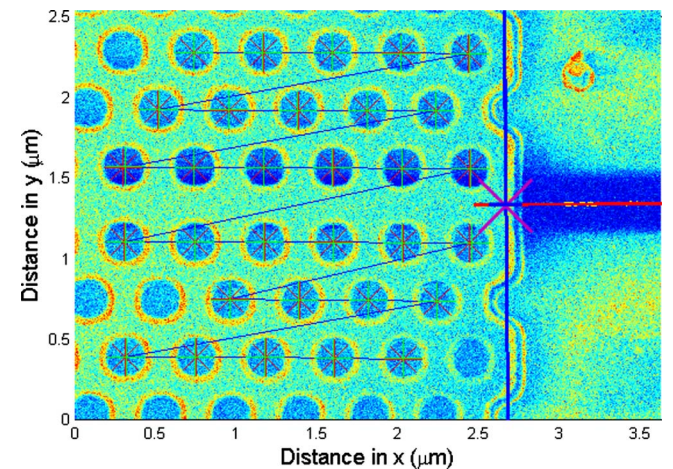

FIG. 8. (Color online) Fitted data for RHS of 4th waveguide, showing the located hole centers (+ crosses) and the fitted lattice (* asterisks joined by continuous zig-zagging line). The cross (×@2.669,1.324) shows the predicted end of the PhC waveguide. The waveguide center is shown as a noisy line and the linear fit (overlaid) predicts the intersection between the center of the waveguide and the PhC facet at the position marked by the darker cross (+@2.669,1.338).
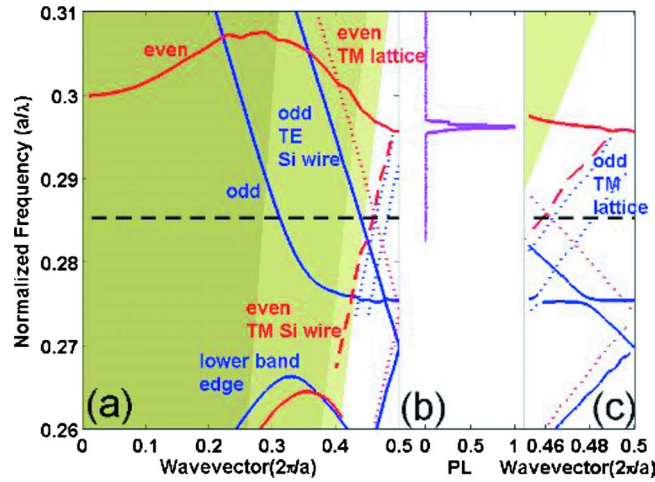
FIG. 9. (Color online) (a) Bandstructure for InP W1–Si wire (326 nm wide) coupled system, for a period, a=455 nm, r/a=0.3, and a vertical separation of 400 nm (BCB layer). The ripple seen in the band structure is mainly due to coupling with lower Q modes. (b) The PL spectrum, coupled out via the Si wire, showing a single peak whose spectral position is consistent with the intersection with the even TE-like W1 mode. (c) Zooming in to the region of k-space close to the K point, we can observe at $(k_x=0.475\times 2\pi/a,\ u=0.275)$ an anticrossing caused by the codirectional coupling between the odd TE-like W1 mode and the odd TE-like Si wire mode.

coupled to a 220×326 nm^2 wide Si wire, situated 400 nm below the PhC waveguide. The space between the two guides is filled with BCB. Compared to the PhC bandstructure represented in Fig. 2(b), we observe in Fig. 9(a) two additional lines representing the TE-like and TM-like Si wire modes, from Fig. 2(a).

One of the consequences of the PhC patterning is that the light emission is expected to be enhanced where the bands become flat, i.e., as v_g approaches zero. For the PhC W1 modes this is the case at the edge of the Brillouin zone for a wave vector $k_x=\pi/a$ (TE-like even mode at u=0.295 and TE-like odd mode at u=0.275 on Fig. 9). The coupling between the PhC and the wire locally modifies the dispersion of their coupled supermodes. Even for small distances between the two waveguide layers the modal couplings are not very apparent in the large scale plot of Fig. 9(a). In order to highlight the coupling points we replot k_x over a narrow range between $0.45-0.5\times2\pi/a$ [see Fig. 9(c)]. In the coupled system we observe at $(k_x=0.475\times 2\pi/a,\ u=0.275)$ an anticrossing caused by the codirectional coupling between the odd TE-like W1 mode and the odd TE-like Si wire mode [see Fig. 9(c)]. At higher frequency the even TE-like W1 mode intersects the even TM-like Si wire mode, very close to the mode edge $(k_x=\pi/a,\ u=0.295)$. The periodic patterning also causes the Si mode to fold at the edge of the Brillouin zone [at $(k_x=\pi/a,\ u=0.27)$ on Fig. 9(c)]. This causes a distributed Bragg-type feedback of the light in the Si wire which results in a low v_g point.

In Fig. 10(a) we plot the coupled bandstructure for a W1 with period, a=405 nm, coupled in the same manner to a 220×326 nm^2 wide Si wire. The shift in period allows us to move these lower frequency features toward the emission bandwidth of the QWs.

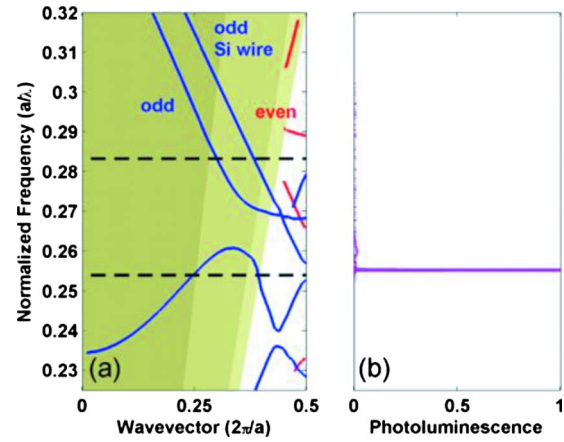We then plot the spectral positions of these various modes for fixed wave vector, $k_x=\pi/a$ as a function of the

FIG. 10. (Color online) (a) Band structure for InP W1-Si wire (326 nm wide) coupled system, for period a=405 nm, r/a=0.3, and a vertical separation of 400 nm (BCB layer). (b) The PL spectrum, coupled out via the Si wire. As the emission is dominated by odd modes, here we only plot the even bands in the region $k_x=0.45-0.5\times2\pi/a$. The dashed black lines denote the emission bandwidth of our QWs.

lattice constant of the PhC (see Fig. 11). On top of this data we plot the measurements from the optical experiments and compare these to the simulation.

B. Optical characterization and discussion

The fabricated samples are explored at room temperature as depicted in Fig. 1 by measuring the light emitted under optical pumping. The samples are surface pumped using either a $10\times$ or a $50\times$ IR long working distance objective. The laser source used for pumping is a Ti:Sa laser operating at a repetition rate of 80 MHz providing 100 fs pump pulses at 810 nm. This near infrared (NIR) pump is absorbed by each of the semiconductor materials, the Si, InP, and InGaAs QWs and their barriers of InGaAsP. The light emission from the III-V PhC is coupled to the Si wire, guided along 3 mm to the shallow etched TE grating couplers and diffracted into a single mode optical fiber (SMF). The SMF is angled at 10° with respect to the surface normal for optimal coupling at
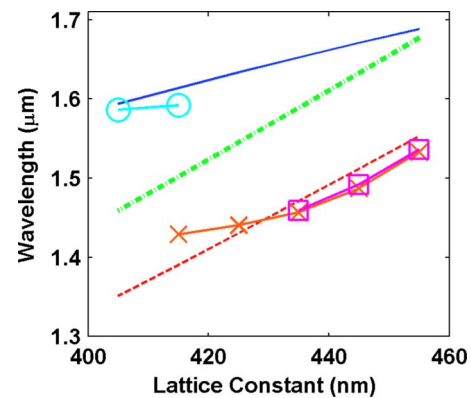

FIG. 11. (Color online) The spectral positions of the various bands at the K point, $k_x=\pi/a$ as a function of PhC period. The top solid line represents the odd TE-like Si wire mode. The middle dot-dashed line denotes the odd TE-like W1 mode. The lower dashed line represents the even TE-like W1 mode. The crosses are surface detected PL. The PL extracted via the Si waveguide from the even mode is marked by squares and the circles correspond to the Si Bragg mode edge.

1.55 μm. The emission is analyzed using a spectrometer equipped with a cooled InGaAs photodiode array. Simultaneously, we detect the light emitted by the surface of the PhC. First, we observe the PL spectra of the samples in order to identify the wavelengths where the emission is enhanced by the low v_g modes of interest. The results of the measurements are shown in the central panel of Figs. 9(b) and 10(b) and their peak wavelengths are summarized in Fig. 11.

In Fig. 11, the wavelength of the observed resonances are plotted as a function of the lattice constant of the PhC for each sample in the range a=405–455 nm in steps of 10 nm and for a fixed r/a ratio of 0.3. The cyan circles on the left shows the measured resonance and corresponds to the odd TE-like Si wire mode folding, detected via the guide [the PL data are also represented in Fig. 10(b)]. The orange crosses at 1430 and 1440 nm are only observed in surface measurement, because these wavelengths fall out with the coupling bandwidth of the grating couplers. The magenta squares show the emission channelled via the Si wire and into the SMF.

Both the odd TE-like (green dot-dashed line) and the even TE-like (red dashed line) PhC modes shift to longer wavelength with increasing period; their dependence upon period show similar gradients. The behavior is different for the spectral position of the folding of the Si wire mode (blue line). The slope of the curve has a shallower gradient because it obeys the relationship $\lambda = 2n_{effSi} \times a$ where n_{effSi} is the effective index of the waveguide mode.

We can see in the central panel of Fig. 9, an enhanced PL emission at about the same frequency as the even TE-like PhC mode (u=0.295). In order to collect the emitted light efficiently via the Si wire we would also expect to be close to a phase matched condition. This even TE-like PhC mode (red solid line on Fig. 9) intersects the even TM-like Si wire mode (dashed red line on Fig. 9), and due to their shared lateral symmetry and the lack of vertical symmetry they can couple. In Fig. 10(b) we see that the PL spectrum, coupled out via the Si wire, shows a single peak whose spectral position is consistent with the K point intersection of the odd TE-like Si wire mode. This is a Bragg effect imposed upon the Si wire modes. This large spectral feature is observed where the Si wire mode is folded by periodicity of the PhC at ($k_x = \pi/a$, u=0.255). Here, the coupling is self-explanatory as the mode belongs to the Si wire system.

At ($k_x = 0.47 \times 2\pi/a$, u=0.275) in Fig. 9(a), we notice that the flat odd TE-like mode of the PhC is intersected by the Si odd TE-like mode which should result in a codirectional coupling. However, in the experimental data in Fig. 11 (425, 435, and 445 nm), no resonances were observed probably due to losses incurred; we merely observe a broadband PL spectrum. This necessitates a deeper study and will be reported in a more exhaustive theoretical paper dealing with the coupling between wire and PhC defect waveguides.[21]

For both even TE-like PhC mode and the odd TE-like Si wire Bragg modes, we observe characteristic lasing behavior. The laser emission captured by the Si waveguide for a =455 nm, 200 periods (91 μm) long PhC waveguide, is plotted in Fig. 12 as a function of the pump power on a log-log scale. We obtain a classical S-shaped curve attesting
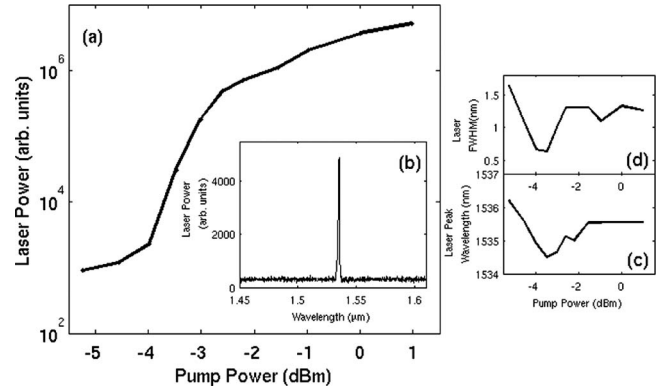

FIG. 12. (a) Laser emission power captured by the Si wire as a function of the pump power in log-log scale (b) Inset: laser spectrum at 1 mW pump power. (c) Laser peak wavelength as a function of pump power (d) Laser FWHM as a function of pump power.

to the transition from a spontaneous to a stimulated emission regime when the incident pump power is greater than 0.4 mW (5 pJ at 80 MHz). The laser peak occurs at 1.54 μm as shown in the inset of Fig. 12. As the pump power increases both the laser peak wavelength and the emission full width at half maximum (FWHM) initially decrease before increasing back up to a constant value with further increase in pump power, which is evidence for carrier clamping above laser threshold. The maximum output power coupled to the SMF is 1.2 fJ (equivalent to 100 nW average power). Note that similar results are also obtained with shorter PhC waveguides (30 periods long ~13 μm).

VI. CONCLUSIONS

We have fabricated a hybrid structure composed of active InP-based PhC defect waveguides coupled to SOI wires using an adhesive bonding technique. Detection of electron beam alignment marks has allowed the registration between the PhC W1 defect waveguide and the Si wire waveguide to be controlled to better than 30 nm.

The light emitted under optical pumping was spectrally analyzed as a function of the lattice constant and the optical pump power. 3D FDTD numerical simulations help us identify the origin of each of the observed emission peaks as low v_g modes in the bandstructures of the coupled guides. Laser operation was observed at room temperature due to the enhanced light-matter interaction provided by the low group velocity modes of the system. We show that the laser emission is coupled to the subjacent SOI waveguides. Wavelength of operation ranged from 1430 to 1595 nm depending on the value of the lattice constant. The laser with a period a =455 nm exhibited a threshold optical pump power of 0.4 mW and was observed to attain an average output power of 100 nW.

In the present work laser emission is obtained under pulsed excitation. Indeed high index contrast BCB bonded membranes exhibit poor thermal dissipation. Technologically the main challenges are to achieve adequate thermal dissipation to enable CW laser operation. The choice of bonding

material has been shown to be a factor[9,10] in this respect. Accurate control of the thickness of the bonding layer is also critical in attaining repeatable coupling.

Several improvements can be conceived to ameliorate the system. One obvious modification has already been implemented which consists of using the optical modes in the system to pump the sample via the Si wire waveguide so that the pump is absorbed more efficiently and uniformly. This constitutes an alternative to the ambitious goal of electrical injection of these lasers; a topic requiring further study to find configurations for contacting without inducing high optical losses. The PhC defect can also be structured to remove higher order modes and also marginally increase the group velocity of the low v_g modes. The coupled bandstructure (for the same 400 nm vertical offset) is much more perturbed for a W0.65 waveguide coupled to a Si wire than for the W1 studied here. These advances are reported in Ref. 22. It would certainly be very interesting to be able to sweep the dispersion of the Si wire mode through the regimes which contra and codirectional couple to the PhC modes. Both lasing and amplification regimes need to be more thoroughly investigated. This periodic high refractive index contrast environment can provide tight confinement (low optical mode volume) in nanocavities and our high precision alignment technique permits us to envisage ultrasmall active components coupled to passive waveguides.

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